



## The Effect of Zeolite Imidazole Framework Geometrical Structure on Carbon Dioxide Permeability for Mixed Matrix Membrane

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### ABSTRACTS

As the trade-off limitation of polymeric membrane between permeability and selectivity is still the biggest concern, the focus has now diverted into the potential of Mixed Matrix Membrane (MMM). Among the fillers used in MMM, the Zeolitic imidazole framework (ZIF) is one of the metal-organic frameworks (MOF) currently being investigated where findings have shown outstanding performances. Thus, this study aims to investigate the effect of fillers geometry on MMM's gas separation performance. The successful synthesis of ZIF-8 and ZIF-L were confirmed using X-Ray Diffraction (XRD) to measure the crystallinity, Scanning Electron Microscopy (SEM) to observe structure and morphology, and Fourier Transform Infrared (FTIR) to assess the functional groups of the ZIF-L and ZIF-8 particles. The gas permeation test shows that at the highest filler loading of 10wt%, ZIF-L MMM gives higher CO<sub>2</sub> permeability (271 GPU) as compared to ZIF-8 MMM (121 GPU).

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## 1. INTRODUCTION

In the past decade, interest has been shifting on the improvement of the performance of polymeric membranes for gas separation due to the trade-off limit (Robenson, 2008). Ever since various types of membranes have been developed including Mixed Matrix Membrane (MMM). MMM comprises of polymer phase and dispersed phase or fillers such as silica particles (Ahn et al., 2008), zeolites (Bastani et al., 2013; Poshusta et al., 2000), metal-organic framework (Basu et al., 2011; Perez et al., 2009), and carbon molecular sieve (CMS) (Vu et al., 2003). The gas transport mechanism in MMM is usually facilitated by a solution-diffusion mechanism. With the incorporation of inorganic fillers, free fractional volume (FFV) is created allowing the gas molecules to pass through the membrane.

Metal-organic frameworks (MOFs) have been widely integrated into membranes for gas separation studies due to their narrow pore windows and flexibility to be modified according to desired structural and chemical features (Vu et al., 2003). Zeolitic imidazole frameworks (ZIFs) in particular have the bright potential to be applied in MMM as they are flexible, chemically, and thermally stable and have been reported to improve the separation performance of polymeric membrane (Melgar et al., 2015). The incorporation of zeolites in polymers has been proven to improve the CO<sub>2</sub>/CH<sub>4</sub> selectivity by 2.5 folds (Duval et al., 1993). As zeolites have high heat of adsorption with molecules with larger dipole moments, CO<sub>2</sub> adsorption is high for zeolites filler. Nordin et al. reported on the improved CO<sub>2</sub>/CH<sub>4</sub> selectivity by up to 72% with the incorporation of 0.5 wt% of modified ZIF-8 (Nordin et al., 2015). Meanwhile, the incorporation of ZIF-L in PEBA MMM has also shown an improved CO<sub>2</sub>/CH<sub>4</sub> selectivity by 5% (Zhu et al., 2019). However, there is still little to no research that focuses on the effect of ZIF's geometrical dimension on MMM separation performance.

Therefore, the objective of this work is to investigate the effect of fillers geometry on the CO<sub>2</sub>/CH<sub>4</sub> separation performance of MMM. 2D ZIF-L and 3D ZIF-8 fillers in PSF dense membrane will be fabricated. ZIF-L and ZIF-8 will be synthesized following the methods established from previous literature (Nasir et al., 2017; Nordin et al., 2014).

## 2. METHODS

### 2.1. Materials

Zinc nitrate hexahydrate (Zn(NO<sub>3</sub>)<sub>2</sub>•6H<sub>2</sub>O), 2-Methylimidazole (2-MeIM), Triethylamine (TEA), Polysulfone (Psf Udel® P-1700), Tetrahydrofuran (THF), Deionized water. All chemicals were used as received from Sigma Aldrich without prior purification.

### 2.2. Zif-L Synthesis

ZIF-L is prepared by dissolving 0.59g of Zn(NO<sub>3</sub>)<sub>2</sub>•6H<sub>2</sub>O (solution A) in 40mL of deionized water and 1.30g of 2-MeIM (solution B) in 40mL of deionized water. The molar ratio between both is kept at 1:8. As solution B is stirred at a constant speed, solution A is added dropwise and the mixture is stirred for 4 hours. The product is then undergone repeated centrifugation at 10000rpm for 20 minutes with three times washing with fresh methanol. The ZIF-L yield is then dried in the oven overnight at 70°C. The ZIF-L particles are then grounded into fine particles before being activated at 100 °C for 12 hours (Nasir et al., 2017).

### 2.3. ZIF-8 Synthesis

ZIF-8 is prepared adopted from previous literature (Nordin et al., 2015). Firstly, 2g of Zn(NO<sub>3</sub>)<sub>2</sub>•6H<sub>2</sub>O (6.72 mmol), 3.312g of 2-MeIM (40.43 mmol), and a predetermined amount of TEA are dissolved in deionized water. The molar ratio is kept at 1:6:500. The mixture is

stirred vigorously for 30 minutes until a cloudy solution is formed. The obtained product is washed several times with deionized water to remove excess reactants and dried in the oven at 60 °C overnight. The ZIF-L particles are undergone an activation process at 100 °C for 12 hours.

#### 2.4. Membrane fabrication

Dense MMM was fabricated at the constant polymer concentration of 15wt%, adopted from the literature (Jusoh *et al.*, 2014). The fillers loading is varied from 2.5 to 10 wt%. **Table 1** shows the membrane abbreviation with each respected filler composition of all fabricated membranes.

**Table 1.** Prepared membrane with its abbreviation and respective filler loading.

Membrane abbreviation	Filler loading (wt% based on the weight of polymer)
ZL-2.5	2.5
ZL-5.0	5.0
ZL-7.5	7.5
ZL-10.0	10.0
Z8-2.5	2.5
Z8-5.0	5.0
Z8-7.5	7.5
Z8-10.0	10.0

#### 2.5. Characterization

X-Ray Diffraction (XRD) analysis was carried out using Siemens D5000 Diffractometer to measure the crystallinity of ZIF-L and ZIF-8 particles. Scanning ranges between 5° to 40° were employed with copper K $\alpha$  ( $\lambda = 0.1541$  nm at 40 kV and 30 mA) was used as the source of radiation. Scanning electron microscopy (SEM, TM3000, Hitachi) was used to observe the membrane structure and morphology. Fourier Transform Infrared (FTIR) spectra in the range of 4000-400 cm<sup>-1</sup> were obtained to test the functional groups of the ZIF-L and ZIF-8 particles.

#### 2.6. Gas Permeation

The gas permeation was conducted using pure CO<sub>2</sub> and CH<sub>4</sub> gas at 2 bar feed pressure and room temperature (25°C). CH<sub>4</sub> was tested first to avoid membrane plasticization by CO<sub>2</sub> and the flow rate of gas permeated was measured using a bubble flow meter. The permeance, P/l (unit GPU), is then calculated using the following equation 1:

$$P/l = \frac{Q_{STP}}{A \cdot \Delta P} \quad (1)$$

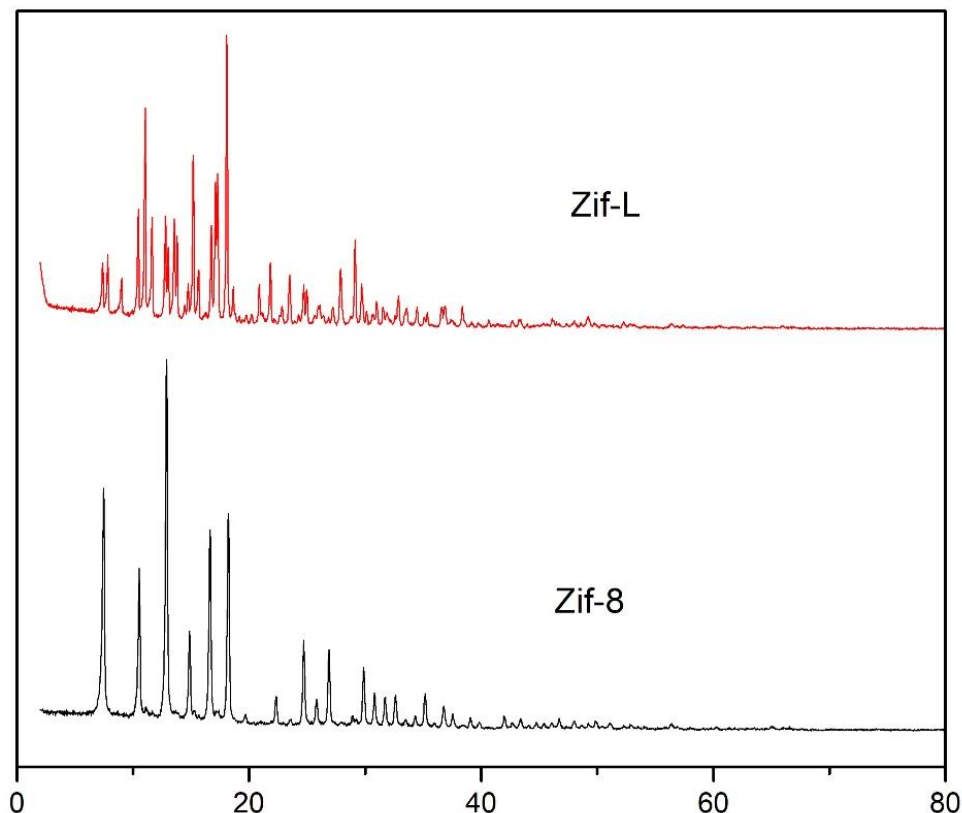
Where  $Q_{STP}$  is the permeate flow rate at standard temperature and pressure.  $A$  is the effective surface area of the membrane film,  $\Delta P$  is the pressure gradient across the membrane. The unit of gas permeance is expressed in GPU (eq. 2).

$$1 \text{ GPU} = 1 \times 10^{-6} \frac{\text{cm}^3(\text{STP})}{\text{sec.cm}^2.\text{cmHg}} \quad (2)$$

### 3. RESULTS AND DISCUSSION

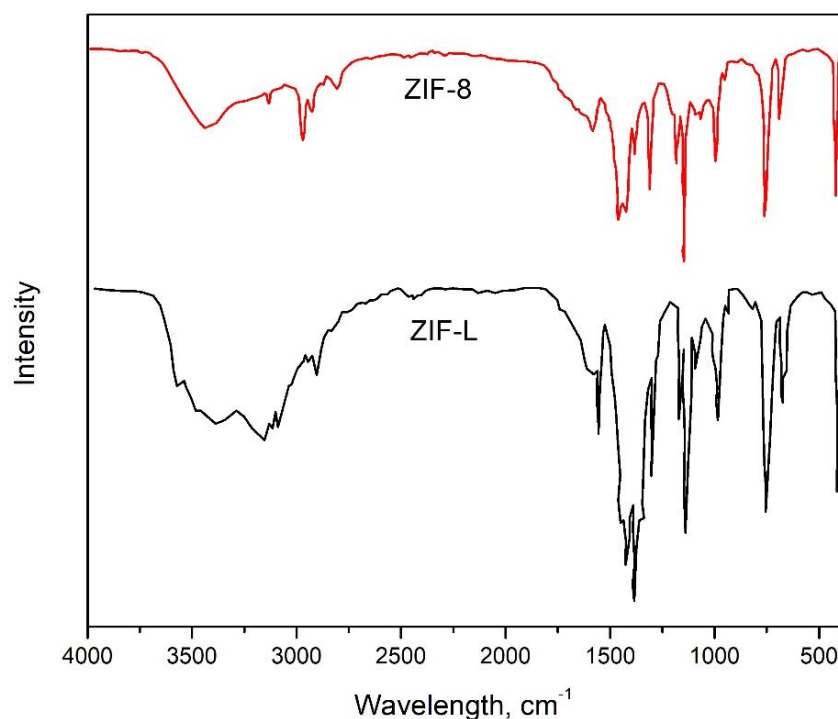
#### 3.1. ZIF-L and ZIF-8 Characterization

**Figure 1** shows the XRD spectra of both ZIF-L and ZIF-8. For ZIF-L, diffraction peaks can be observed at {110}, {200}, {211}, {220}, {310}, and {220} planes which were observed at  $2\theta = 7.40^\circ$ ,  $10.47^\circ$ ,  $12.78^\circ$ ,  $14.80^\circ$ ,  $16.74^\circ$  and  $18.08^\circ$ , respectively. Meanwhile, for prepare ZIF-8 particles, strong peaks at  $2\theta = 7.30^\circ$ ,  $10.35^\circ$ ,  $12.70^\circ$ ,  $14.80^\circ$ ,  $16.40^\circ$  and  $18.00^\circ$ , which corresponds to the planes at {110}, {200}, {211}, {220}, {310}, and {222}, respectively. Both results show that ZIF-L and ZIF-L were successfully synthesized and shows high crystallinity.



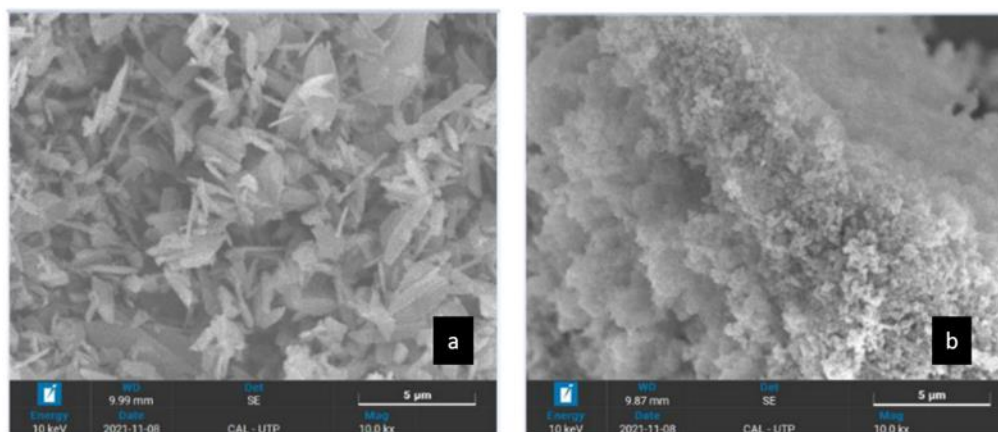
**Figure 1.** XRD spectra of ZIF-L and ZIF-8.

FTIR analysis was conducted to study the functional group of both ZIF-L and ZIF-8 particles and the spectra are plotted in **Figure 2**. ZIF-8 functional groups were defined in the peak band at  $3433.02\text{ cm}^{-1}$  which attributes to the N-H stretching vibration. Both peaks at  $3134.84\text{ cm}^{-1}$  and  $2971.92\text{ cm}^{-1}$  can also be attributed to the aromatic and aliphatic C-H asymmetric stretching vibrations, respectively while the peak at  $1585\text{ cm}^{-1}$  was associated with the C=N stretch vibration. Meanwhile, bands at  $1310.19\text{--}1458.45\text{ cm}^{-1}$  were for the entire ring stretching, and the signal at  $1146\text{ cm}^{-1}$  is for aromatic C-N stretching mode. The peaks at  $996$  and  $759\text{ cm}^{-1}$  were for C-N bending vibration and C-H bending mode, respectively. For ZIF-L, corresponding characteristic peaks can be observed at  $1600\text{ cm}^{-1}$  which represents the C=N stretch in the imidazole ring, while at  $1300\text{ cm}^{-1}$  associated with the C-N stretching band.



**Figure 2.** FTIR spectra of ZIF-L and ZIF-8

SEM analysis was also conducted to observe the morphology of the prepared ZIF-L and ZIF-8 particles at a magnification of 10kx and the result is shown in **Figure 3**. From the result, ZIF-L shows the leaf-like shape while ZIF-8 shows a hexagonal-like shape in line with literature (Khan *et al.*, 2018).

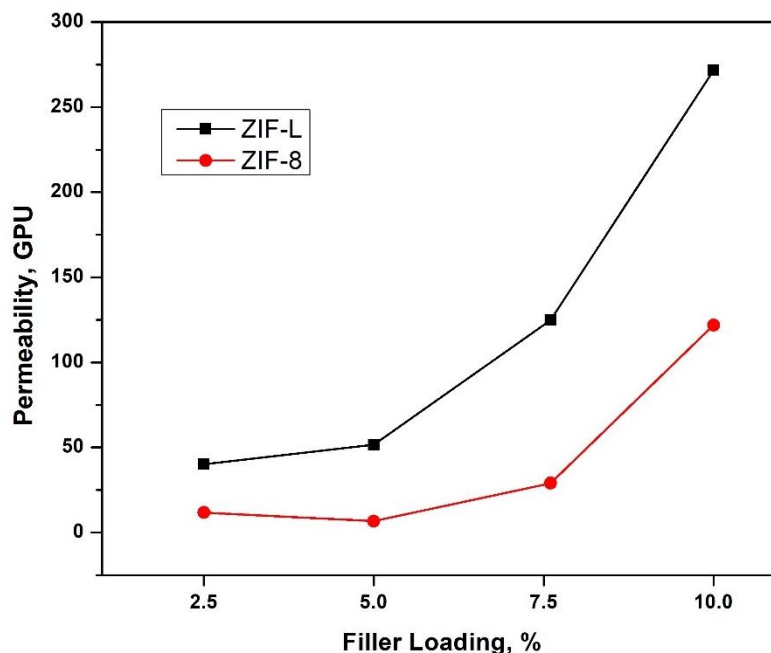


**Figure 3.** SEM images of a) ZIF-L and b) ZIF-8 at a magnification of 10kx.

### 3.2. Gas

**Figure 4** shows the CO<sub>2</sub> permeability of all prepared ZIF-L and ZIF-8 MMM at different filler loading. Firstly, for ZIF-L MMM, ZIFL-2.5 shows CO<sub>2</sub> permeability of 40.18 GPU. With increasing loading, ZIFL-5.0, ZIFL-7.5, ZIFL-10.0 shows a significant increase of 27.5%, 210.0%, 577.5%, respectively. Meanwhile, for ZIF-8 MMM, the CO<sub>2</sub> permeability of ZIF8-2.5 starts at 11.71 GPU. With increasing loading to ZIF8-5.0, the CO<sub>2</sub> permeability drops a little by 1.75 folds but increases with ZIF8-7.5 and ZIF8-10 by 2.47 folds and 10.41 folds, respectively. Overall, for all filler loading, ZIF-L MMM displays a higher CO<sub>2</sub> permeability compared to ZIF-

8 MMM. By configuration, ZIF-8 promotes higher CO<sub>2</sub> permeation due to having pore openings of 3.4Å equivalent to the kinetic diameter of CO<sub>2</sub>. The decrease in CO<sub>2</sub> permeability in ZIF-8 MMM is probably due to (1) polymer chain rigidification which would hinder the CO<sub>2</sub> permeation across the membrane hence low permeability or (2) due to pore blockage induced by the polymer during membrane fabrication. Therefore, as an initial observation, it can be postulated that ZIF-L MMM gives higher CO<sub>2</sub> permeability compared to ZIF-8 MMM.



**Figure 4.** CO<sub>2</sub> permeability of ZIF-L and ZIF-8 MMM at different filler loading.

#### 4. CONCLUSION

The effects of MOF's geometrical structure on the gas separation performance of MMM are investigated. PSF membrane incorporated with different loadings of ZIF-L and ZIF-8 particles were successfully fabricated as indicated by the characteristic peaks shown from XRD and FTIR analysis. The CO<sub>2</sub> permeability test shows that ZIF-L MMM gives higher CO<sub>2</sub> permeability as compared to ZIF-8 MMM. At the highest filler loading of 10wt%, ZIF-L MMM gives an almost 123% increment in CO<sub>2</sub> permeability compared to that of ZIF-8 MMM. Although the dimensional structure of ZIF-L and ZIF-8 MMM has shown significant findings in the CO<sub>2</sub> permeability, further study on the filler's distribution across the membrane and its effect on gas solubility and diffusivity would be beneficial to further understand this topic.

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#### 6. AUTHORS' NOTE

The authors declare that there is no conflict of interest regarding the publication of this article. Authors confirmed that the paper was free of plagiarism.

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